

RESEARCH ARTICLE	Properties and perspective applications of copper sullide nano-materials: a review	
Ofeliya Balayeva	Dr	
	Baku State University	
	Azerbaijan	
	Email: educ25000@gmail.com	
Doi Serial	https://doi.org/10.56334/sei/8.6.66	
Keywords	Copper sulfide, nanostructures, optical properties, structural properties, energy stor- age and conversion (ESC) devices.	
Abstract		
In this review we could get more information about nanostructured copper sulfide, its properties and applications in many fields. The key reaction parameters that influence the nucleation and growth as well as the size, shape, composition, and structure are carefully discussed.		
Citation		
Balayeva O. (2025). Properties and perspective applications of copper sullide nano-materials: a review. <i>Science, Education and Innovations in the Context of Modern Problems</i> , 8(6), 603-614; doi:10.56352/sei/8.6.66. <a href="https://imcra-az.org/archive/364-science-education-and-innovations-in-the-context-of-modern-problems-issue-6-volviii-2025.html">https://imcra-az.org/archive/364-science-education-and-innovations-in-the-context-of-modern-problems-issue-6-volviii-2025.html</a>		
Licensed		
© 2025 The Author(s). Published by Science, Education and Innovations in the context of modern problems (SEI) by IMCRA - International Meetings and Journals Research Association (Azerbaijan). This is an open access article under the CC BY license ( <a href="http://creativecommons.org/licenses/by/4.0/">http://creativecommons.org/licenses/by/4.0/</a> ). PACS: 61.46. <a href="#">Science, Education and Innovations in the Context of Modern Problems, Issue 6, Vol.VIII, 2025</a>		
Received: 05.02.2025	Accepted: 29.04.2025	Published: 28.05.2025 (available online)

## Introduction

Semiconductor nanocrystals are always metastable phases, and they should be passivated by ligands to avoid aggregation, though it is hard to meet this goal by the above mentioned solid-solid or solid-gas reactions. Nanocrystalline copper sulfide exhibits at least five stable phases with different Cu:S molar ratios. These phases include covellite (CuS), anilite (Cu<sub>1.75</sub>S), digenite (Cu<sub>1.8</sub>S), djurite (Cu<sub>1.95</sub>S), and chalcocite (Cu<sub>2</sub>S) [1]. Copper sulfide grows with 0-D (quantum dots [2,3], 1-D (nanotubes [4,5]; nanowires [6-8]; nanorods [9,10]; nanoneedles [11,12]), 2-D (nanoplates [13-15]; nanoribbons [16,17];

nanodisks [17]; nanosheets [11,18]) and 3-D (nanocubes [19], nanospheres [20-23], cages [24, 25]. Therefore, there is a strong tendency to minimize the total surface energy. Owing to its unique electronic, optical and other physical and chemical properties [26, 27] and wide potential applications in many fields [28], copper monosulfide (CuS) semiconductor has attracted much attention.

Among these various morphologies, hollow spheres with nano- or micro-sized dimensions gain extensive attentions because of its prospective applications in catalysis, drug delivery, light fillers, chemical storage and photonic crystals [29-31].

In this review we could get more information about nanostructured copper sulfide, its properties and applications in many fields. The key reaction parameters that influence the nucleation and growth as well as the size, shape, composition, and structure are carefully discussed.

**1. Properties of Copper Sulfide Nanomaterials.** Owing to its unique electronic, optical and other physical and chemical properties [26, 27] and wide potential applications in many fields [28], copper monosulfide (CuS) semiconductor has attracted much attention. In this section, we will selectively summarize the main and general properties

of copper sulfide nanomaterials.

**1.1. Structural properties.** The crystal structure of covellite is hexagonal (space group  $P6_3/mmc$ ,  $a$

$= 380$  pm,  $c = 1640$  pm) with 6 CuS formula units in the unit cell. Of the 6 Cu atoms 4 are tetrahedrally coordinated and 2 are triangularly coordinated by S atoms. Four of the 6 S atoms form S groups. In accordance with this, the S (2p) X-ray photoelectron spectrum of CuS shows a doublet structure corresponding to the two different coordinations (octahedrally and triangularly bipyramidal) of S atoms [32–34]. The electron-diffraction results presented can, however, be readily interpreted as a mixture of hexagonal chalcocite and high digenite [35]. Kimihiko Okamoto et al. [36] observed what they denoted as “indistinct polymorphic phase transitions” in the temperature region 90 to 310 K. Such transitions have not been confirmed by other investigators. The heat capacity was extrapolated to the decomposition temperature of covellite, 780 K [34].

Both the precursor molar concentration, reaction time and the reaction temperature are crucial for obtaining pure CuS. CuS and Cu<sub>2</sub>S hydrothermally synthesized at (a) 150 °C, (b) 180 °C, and (c) 250 °C with varied precursor [Cu(NO<sub>3</sub>)<sub>2</sub>: Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub>] molar ratios [13]. The XRD patterns of the product obtained with a [Cu(NO<sub>3</sub>)<sub>2</sub>: Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub>] = 1 : 2 molar ratio at 150 and 180 °C are found to be similar and well-matched to pure hexagonal CuS. However, the product obtained at 250 °C shows two additional diffraction peaks, which are assigned to Cu<sub>2</sub>S along with the major hexagonal CuS diffraction peaks. The relatively high pressure generated during the high temperature (250 °C) hydrothermal synthesis leads to the desulfurization of some CuS to Cu<sub>2</sub>S. Further increasing the thiosulfate concentration, i.e., 1 : 6 molar ratio of Cu(NO<sub>3</sub>)<sub>2</sub> to Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub>, they obtained CuS along with sulfur at all three reaction temperatures.

This is obviously due to the excess amount of thiosulfate in the reaction medium [13]. An optimum precursor [Cu(NO<sub>3</sub>)<sub>2</sub>: Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub>] molar ratio of 1 : 2 is suitable for synthesizing pure CuS in the temperature range of 150–180 °C, whereas with a molar ratio of 1 : 4, the synthesis temperature should be higher than 200 °C, so the excess sulfur formed could be degraded. However, Changhua An et al. reported pure CuS with a 1 : 6 precursor molar ratio of copper acetate and sodium thiosulfate in ethylene glycol solvent and sodium oleate and hexadecylamine as surfactant at 180 °C [37]. This suggests that, in aqueous medium, is suitable for obtaining pure CuS in a shorter reaction time with a lower thiosulfate precursor concentration.

**1.2. Optical properties.** In the Cu–S system, it has many known stable phases from chalcocite (Cu<sub>2</sub>S) to sulfur-rich covellite (CuS). Each stable phase has its own characteristic optical property. UV–visible absorption spectroscopy is a useful technique to monitor the optical properties of the nanomaterials. Semiconductor nano-crystallites are known to have an absorption edge, which is shifted with respect to the bulk material towards shorter wavelengths. This blue shift is thus taken as an indication of the presence of such nanocrystals, and can be used to evaluate the crystallite size [38]. In a previous study, CuS nanowires [6] have been reported by hydrothermal method. Optical absorption of it is reported; the product obtained at lower temperature (120–175 °C) with CuS nanowire shows a small absorption peak around 400 nm and has broad absorption in the near-IR.

For some Cu–S system nanoparticles [39] they point out to have two phases (CuS and Cu<sub>2</sub>S), the presence of a characteristic broad absorption band in UV region for Cu<sub>2</sub>S and the other in near-IR region for covellite copper sulfide (CuS).

Such a copper sulfide group (Cu<sub>x</sub>S) involves various compounds, Cu<sub>2</sub>S (chalcocite), Cu<sub>1.96</sub>S (djur-leite), Cu<sub>1.8</sub>S (digenite) and CuS (covellite). The electronic structures depend on their stoichiometries, i.e., the copper deficiency. Cu<sub>1.96</sub>S, Cu<sub>1.8</sub>S and CuS are of a direct band gap type, while Cu<sub>2</sub>S is of an indirect band gap type. The energy band gap varies from 1.2 eV for  $x = 2$  and 1.5 eV for  $x = 1.8$  to 2 eV for  $x = 1$  [40]. Since their optical properties also depend on the size and shape of copper sulfide nanocrystals, it is important to control their morphology and composition for application to optoelectrical devices.

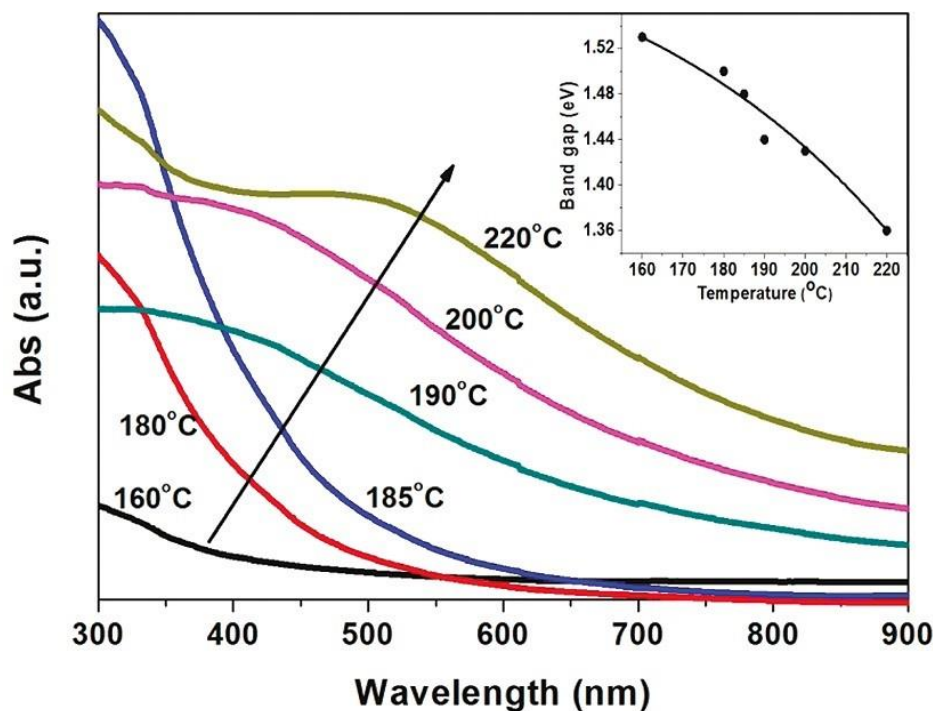
For the CuS nanoparticles prepared at (a) 150 °C, (b) 180 °C, and (c) 250 °C with a precursor molar ratio of 1 : 2 [Cu(NO<sub>3</sub>)<sub>2</sub>: Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub>] [13] a broad peak is observed in the region of 600–650 nm for the CuS prepared at 150 and 180 °C, whereas the CuS prepared at 250 °C shows a peak at ~675 nm. The measured bandgap is 1.75 eV for CuS prepared at 150 °C whereas it is 1.68 eV for CuS prepared at 180 and 250 °C suggesting potential applications in optoelectronic devices.

The hot injection method used 1-octadecene and tri-*n*-octylphosphine solution medium, cuprous acetate and dodecanethiol precursors to synthesize Cu<sub>2</sub>S nanodisks at 160–220 °C, and the size and aspect ratio could be controlled by changing refluxing temperature and time [23]. The band gap of the products plotted as a function of reaction temperature shows a gradual decrease from 1.53 to 1.36 eV with increasing temperature. The decrease of the band gap is ascribed to the increased size of Cu<sub>2</sub>S nanocrystals at higher reaction temperatures (Fig. 1).

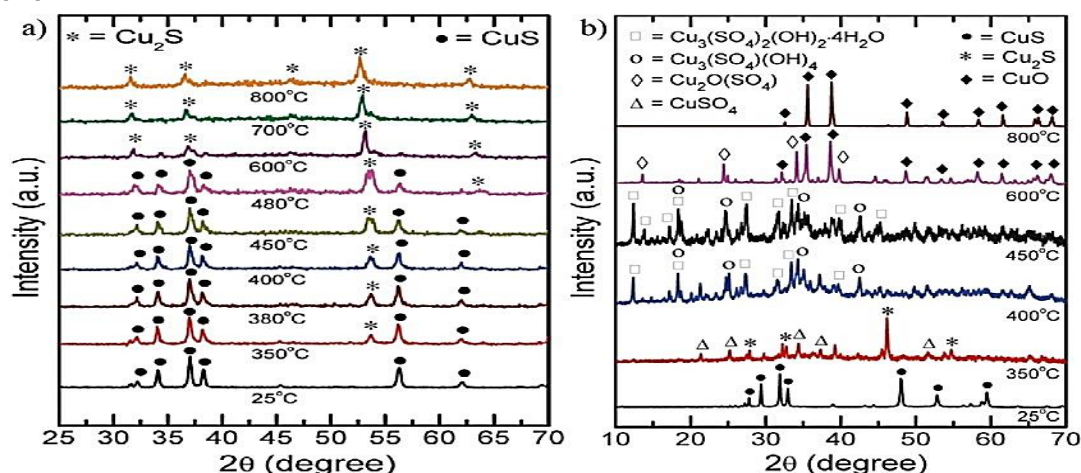
The growth of Cu<sub>2</sub>S nanocrystals has been monitored by characterizing samples collected at different reaction times by using TEM and UV-vis absorption spectroscopy. The diameter/thickness of Cu<sub>2</sub>S disks changed from 13/7.5 nm to

18.2/9.6 nm gradually when the reaction time extended from 60 to 300 min [23]. After the relatively fast nucleation at around 15 min, the slow growth of Cu<sub>2</sub>S nanocrystals gradually strengthened the absorption of the nanocrystals, leading to a red shift of the peak as the reaction proceeds. The relatively fast nucleation and slow growth processes are believed to contribute to the high uniformity of the nanocrystals [41]. One can notice slight broadening in size for samples heated for along time (e.g., 540 min) because of the Ostwald ripening process. The final product after 540 min of reaction shows a very broad absorption spanning from UV to near IR which may make this material a good candidate for photovoltaic devices.

**1.3. Thermal properties.** The presence and degradation of elemental sulfur formed with CuS was characterized by thermal analysis [Cu(NO<sub>3</sub>)<sub>2</sub> and Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub>] molar ratios [13]. They were heated CuS at different temperatures in a vacuum



and/or in air and found different results (figure 2). The residue obtained upon calcining pure CuS at 350 °C shows the formation of Cu<sub>2</sub>S, which is in accordance with the TG analysis. However, the pure CuS product calcined at 400 and 450 °C produced copper sulfate hydroxide hydrate and copper sulfate hydroxide, as shown in Fig.2. According to fig. 2, the oxygen present for formation of CuO at 800 °C. Upon heating the pristine CuS to 350°C in a vacuum, a Cu<sub>2</sub>S phase emerges [13].



**Figure 1.** UV-Vis-NIR absorption spectra of Cu<sub>2</sub>S nanocrystals obtained at different reaction temperatures; the inset shows variation of band gap as a function of reaction temperature. [23]

**Fig. 2.** (a) XRD patterns of CuS heated at different temperatures in a vacuum. High temperature heating induced the phase

*change of CuS to Cu<sub>1.81</sub>S. (b) XRD patterns from the residues of pure CuS calcined at different temperatures in air.*

This is in accordance with the thermal analysis, which shows the desulfurization of CuS to Cu<sub>1.81</sub>S. Upon further increasing the temperature up to 480 °C, the intensity from Cu<sub>1.81</sub>S diffraction features is found to increase, which indicates more desulfurization of CuS to Cu<sub>1.81</sub>S. It is also important to note that the most intense diffraction feature of Cu<sub>1.81</sub>S, at 2θ of ~54°, is found to be shifted to a lower 2θ with an increase in the temperature of the sample. This is due to the lattice expansion of the crystal at higher temperatures [13].

Liqiu Wang et al. [42] apply the chemical solution method to synthesize CuS/Cu<sub>1.81</sub>S nanofluids and experimentally measure their thermal conductivity. The measured thermal conductivity shows that the presence of nanoparticles can either upgrade or downgrade fluid conductivity, a phenomenon predicted by the recent thermal wave theory of nanofluids. The presence of CuS/Cu<sub>1.81</sub>S nanoparticles can either enhance or weaken fluid heat conduction so that the conductivity ratio varies from 0.82 (smaller than 1) to

1.21 (larger than 1).

**1.4. Electrical properties/ Photoelectric and Thermoelectric properties.** Electrochemical capacitors (ECs), also called supercapacitors, are attracting particular attention owing to their high power density, rapid charging/discharging rate, and long cycle life. ECs are divided into two types, depending on their underlying energy storage mechanism: electrical double-layer capacitors (EDLCs) and pseudo capacitors [43]. Whereas EDLCs store electrical energy by the electrostatic accumulation of charges in an electric double-layer close electrode/electrolyte interfaces, pseudo capacitors exploit reversible Faradaic reactions that occur at the electrode surface, and so have a much higher specific capacitance [44]. Coppersulfide is one of the promising cathode materials of lithium ion batteries (LIBs) because of its advantages of a high theoretical capacity 560 mAh g<sup>-1</sup>, flat discharge curves and a good electrical conductivity (10<sup>-3</sup> S cm<sup>-1</sup>) [45]. As an important semiconductor material, copper sulfides exhibit their unique physical and chemical properties. For instance, copper sulfide (covellite) shows a metal-like electrical conductivity

[46] and transforms into a superconductor at 1.6 K [47]. Anup Mondal et al [48] reported a chemical route for the deposition of nanocrystalline thin films of CuS, using aqueous solutions of Cu(CH<sub>3</sub>COO)<sub>2</sub>, SC(NH<sub>2</sub>)<sub>2</sub> and N(CH<sub>2</sub>CH<sub>2</sub>OH)<sub>3</sub> [triethanolamine, i.e. TEA] in proper concentrations and ratios. For electrical characterizations, the CuS films were deposited on fluorine doped tin oxide (SnO<sub>2</sub>:F) coated transparent conducting oxide (TCO) glass substrates and the d.c. I-V measurements within the range of -3.0 V to +3.0 V were carried out. The upper electrical contact to the film was made with carbon paste of 0.09 cm<sup>2</sup> area, while, TCO itself acted as the bottom contact [48].

A. P. Alivisatos et al. fabricated a hetero-junction of CuS on indium-tin-oxide glass and flexible plastic substrates by spin coating to study its photovoltaic performance [49]. Jeunghee Park et al. reported polymer solar cells using CuS-MWCNTs nanocrystals [50]. For the photovoltaic application, it is important to fabricate the materials as thin film forms. Copper sulfide thin films can be obtained directly by electrochemical or atomic layer deposition processes [51] or drop-casted to fabricate a thin film. Glass, polyester, and metals are generally used as the substrate for depositing thin films of CuS [52].

**1.5. Field-emission (FE) properties.** As compared with conventional technologies, nanostructures have many advantages such as faster device turn-on time, sustainability and compactness [53]. Recent progress in the research on FE and stimulated emission properties of 1D wide band-gap semiconducting metal sulfide nanostructures has resulted in an important increase in the current density and decrease in the turn-on voltage [54]. Hierarchical microrings and porous microrings were synthesized through a facile solvothermal method by Ke Yu et al. [55]. In some recent researches, FE properties of Cu<sub>1.81</sub>S nanostructures with different morphologies were reported [56].

Ke Yu et al. investigated the FE properties of self-mosaic Cu<sub>1.81</sub>S nanoflowers and compared them to those of the original self-assembled nanoflower sample [57]. For FE measurement, self-mosaic and original self-assembled Cu<sub>1.81</sub>S samples were painted on Si substrates by a screen-printing method respectively. FE properties were measured under a high vacuum level of about 2×10<sup>-5</sup> Pa at room temperature. The emission area was 1cm×1cm. Generally, FE mainly depends on the emitter geometry [58]. The Cu<sub>1.81</sub>S nanostructures with thin edges on the surface are supposed to have good FE properties. The emission current density exponentially increases with the increase of applied field. The turn-on field and threshold field for original and self-mosaic Cu<sub>1.81</sub>S samples are about 5.2V/μm and 8.4V/μm, 4.4V/μm and 7.6V/μm, respectively. It is clear that the FE property for the self-mosaic Cu<sub>1.81</sub>S sample is better than the other one. In the case of their self-mosaic Cu<sub>1.81</sub>S sample, the applied primary field gets enhanced on the original nanosheets, which successively act as a secondary field for the small self-mosaic nanosheets, thereby enhancing the local electric field. For these reasons, self-mosaic Cu<sub>1.81</sub>S sample shows better FE properties. Three flower-like and one rod-shape cubic Cu<sub>1.81</sub>S nanostructures were successfully synthesized from a facile hydrothermal method [59]. Field emission (FE) properties of these nanostructures results indicated that the tetragonal Cu<sub>1.81</sub>S nanorods had excellent field emission performance with turn-on field of ~2.2V μm<sup>-1</sup>, threshold field of ~5.1 V μm<sup>-1</sup> and enhancement factor of 1532. It showed that the tetragonal Cu<sub>1.81</sub>S nanostructures were competitive material in FE applications. The Cu<sub>1.81</sub>S nanoparticles have lower work function than Cu<sub>2</sub>S. Therefore, it is

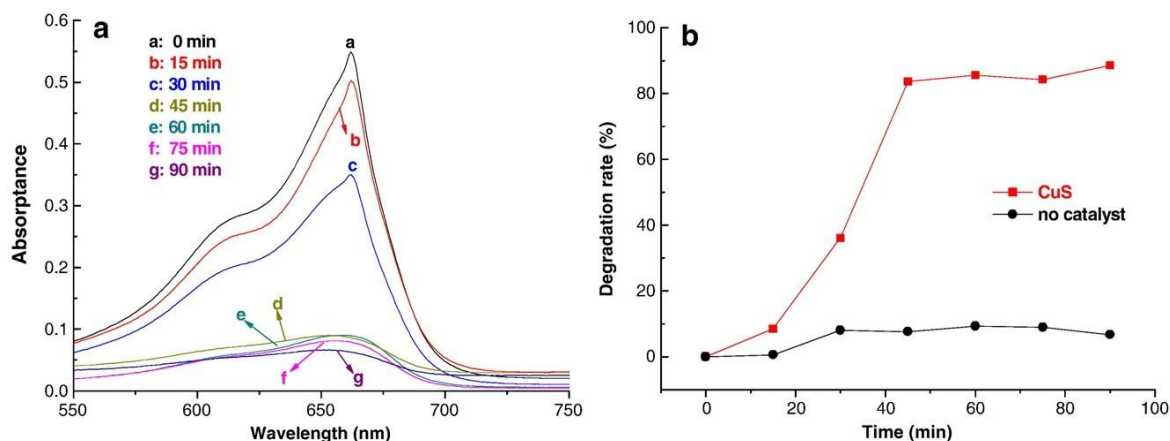


reasonable that electrons can be emitted more easily from these sharp edges of  $\text{Cu}_{1.81}\text{S}$ . Moreover, the result shows that nanorods own better emitter geometry than nanoflowers and the field emission property for the  $\text{Cu}_{1.81}\text{S}$  nanorods compares favorably with that of the multiwall CNTs [60].

**1.6. Photocatalytic activity.** Nanostructured semiconductors have faster electron migration and charge generation rate due to their higher surface area-to-volume ratio than the bulk [61]. Semiconductor nanomaterials for photocatalytic degradation of organic contaminants are promising materials in environmental protection nanotechnology [54]. The development of new visible light-active photocatalysts is one of the most important subjects in photocatalysis research. Doping of metal ions into wide band gap metal sulfide semiconductor nanomaterials has been used to prepare photocatalysts [62].

The photocatalytic performance of the hierarchical CuS structures on the oxidation of methylene blue (MB) dye has been studied in the presence of hydrogen peroxide [63]. The catalytic reaction was conducted under natural light. UV-Vis spectra were applied to demonstrate the photocatalytic degradation activity of MB. The intensity absorption peak at 662 nm of MB decreased gradually with the extension of time, indicating the photocatalytic degradation of MB. About 90% of the MB was degraded after 90 min (fig. 3). In their work, the hierarchical CuS structures have a lot of unfolded nanoplates that could absorb more photons to produce electron-hole pairs under natural light, and the deep and capacious pores enable the hierarchical structures to be exposed to the MB solution sufficiently. Furthermore, the nano-size of plates could reduce the radiationless recombination of electron-hole pairs [64], which is also in favour of the photocatalysis of MB.

Currently, the studies on the photocatalytic properties of CuS nanostructures are seldom reported. G.C.Guo et al. [65] reported the photocatalytic properties of CuS nanoflowers on Rhodamine B and the degradation rate is close to 90% after 2 h halogen lamp irradiation.



**Fig. 3.** (a) Absorption spectra of MB aqueous solutions in the presence of the hierarchical CuS structures. (b) Degradation rate of MB at different intervals [63].

Andrews Nirmala Grace et al. explained the mechanism of photocatalytic reaction in their work [66]. They prepared CuS nanostructures by hydrothermal route using copper nitrate and thiourea as copper and sulfur precursors. The photocatalytic activity was evaluated by the decolorization of methylene blue (MB) dye under visible-light irradiation and results showed that 87% of the dye was degraded. The photocatalytic degradation of methylene blue was monitored under visible light irradiation and the measurement was taken using a UV-Vis spectrophotometer [66]. For the adsorption of dye molecules, electron-hole pair plays an important factor. Initially the dye molecule gets adsorbed on the surface of the catalyst. When CuS photocatalyst absorbs light from the visible region, it will produce pairs of electrons and holes, which further react with dye molecules to oxidize the dye pollutant into non-toxic products. Under visible light radiation, an electron from the valence band gets excited to conduction band to form electron-hole pairs. The electrons from the conduction band are transferred to the catalyst surface reducing  $\text{O}_2$  to  $\text{O}_2^-$  [66]. On the other hand, the holes in the valence band react with  $\text{H}_2\text{O}$  on the surface of the catalyst to form hydroxyl radicals. These radicals show high oxidizing ability reacting with organic dyes to oxidize it. Thus the as-prepared CuS catalysts are highly promising materials for photocatalytic applications towards dye degradation.

High quality CuS and CuS/ZnS core/shell nanocrystals (NCs) were synthesized in a large quantity using a facile hydrothermal method at low temperatures of 60 °C and evaluated in the photodegradation of Rhodamine B (RhB) under visible light irradiation [67]. Photocatalytic activity originates from photo-generated holes in the narrow bandgap CuS, with encapsulation by large bandgap ZnS layers used to form the core/shell structure that improves the resistance of CuS towards photocorrosion. Such CuS/ZnS core/shell structures exhibit much higher photocatalytic activity than CuS or

ZnS NCs alone under visible light illumination, and is attributed to higher charge separation rates for the photo-generated carriers in the core/shell structure [67]. The high photodegradation efficiency of CuS/ZnS core/shell catalysts can be explained by the formation of type-II structure, which promotes the separation of the photoexcited electrons and holes which contribute to the photocatalytic decomposition reaction of RhB. The stability of the CuS/ZnS core/shell particles and impact of the ZnS shell in resisting photocorrosion was verified by exploring the reusability of the photocatalyst. After five photodegradation recycles of RhB, the CuS/ZnS core/shell catalysts did not display any significant loss in the photocatalytic activity, confirming that CuS/ZnS core/shell NCs were good photocatalytic materials. These materials open up exciting opportunities for visible light initiated photocatalytic depollution and subsequent studies will evaluate the application of these catalysts under continuous operation [67].

**2. Perspective applications.** Nanostructured copper sulfide exhibit size and shape dependent properties and have potential applications in various fields, such as materials science; biomedical science [4,68]; electronics [69,70]; optics [71]; energy storage [20], electrochemistry [69,70] and so on. As a typical transitional metal chalcogenide, copper sulfide is a well known p-type semiconductor that has potential in applications of solar controller, solar radiation absorber and lithium ion batteries [72,73].

Synthetic colors are widely used in food industries; while some of them may generate [74] hazards and problem to human health at excess level [74]. Reza Sahraei et al. [75] were synthesized copper sulfide nanoparticles loaded on activated carbon (CuS-NP-AC) by novel, low cost and green approach was used for the removal of sunset yellow (SY) from aqueous solutions. Another resembling work, Mehro-rang Ghaedi et al. synthesized CuS-NP-AC adsorbent was utilized to removal of AO (Auramine O (4,4'- dimethylaminobenzophenonimide) and SO (Safranin O -2,8-dimethyl-3,7-diaminophenazine) dye from aqueous medium in presence of ultrasound radiation [76]. It was found that application of ultrasound leads to significant enhancement in extent of adsorption and removal percentage of AO and SO from aqueous solutions. Combination of ultrasound with CuS-NP-AC increased the dye removal percentage (more than 99%) by using small amount of adsorbent (0.06 g) in short time (7 min). The proposed method has good potential for removal of dyes from waste water compared to several other adsorbents.

Special rose-like copper (I) sulfide nanostructure (Cu<sub>2</sub>S NRS) made up of thickness nanoplates were successfully synthesized on copper rod by a simple method can be used as integrated electrode (called as Cu<sub>2</sub>S NRS/CRIE) to nonenzymatic detect the cholesterol (CT) [16]. In the past, many works were devote to fabricate enzyme-based sensors because it's high sensitivity and good selectivity [77,78], but the activity of the enzyme decreases with use and the enzyme is easily denatured during its immobilization procedure due to its intrinsic instability and the activity of the enzyme is liable to be affected by pH, temperature and toxic chemicals. Moreover, the reproducibility of enzyme-based sensors is not very good and needs to be further improved [79]. In the [16] work results show the Cu<sub>2</sub>S NRS/CRIE displayed excellent performance, a wide linear range from 0.01 mM to 6.8 mM for the detection of cholesterol with a low detection limit of 0.1  $\mu\text{M}$  (S/N = 3).

Glucose detection is extremely important for diabetes clinical detection and therapy [80]. In recent years, variety of Cu based nanomaterials have been reported toward the development of nonenzymatic glucose sensors [81], but the nonenzymatic glucose sensor based on hollow structured CuS was rarely reported. There are several methods reported for the synthesis of the CuS nanotubes such as oleic acid/water microemulsion method and microwave assisted Cu complex transformation method [82]. In these two reports, the as-prepared CuS NTs could be used for nonenzymatic glucose sensing at low glucose concentration with high sensitivity [4].

## 2.1. Application in energy storage and conversion (ESC) devices.

**2.1.1. Batteries.** Copper sulfides have been investigated as battery active materials for lithium-ion batteries (LIBs) for several decades. Cu<sub>2</sub>S nanocrystals are widely used in LIBs. Owing to the increasing demand of energy and shifting to the renewable energy resources, (LIBs) have been considered as the most promising alternative and green technology for energy storage applied in hybrid electric vehicles (HEVs), plug-in hybrid electric vehicles (PHEVs), and other electric utilities [83]. Xuelin Yang et al. reported the fabrication of porous Cu<sub>2</sub>S nanostructure that directly grows on Cu foam via a novel low temperature dry thermal sulfuration method [84]. The electrochemical performance of the Cu<sub>2</sub>S/Cu as cathode for Li-ion battery was studied by charge/discharge test and cyclic voltammetry (CV) measurement. The results indicate that the Cu<sub>2</sub>S/Cu cathode exhibits excellent cycle stability and rate capability. When applying a charge/discharge rate of 0.25 C, it delivers initial discharge and charge capacity of 0.74 and

0.40 mAh cm<sup>-2</sup>, respectively. After 100 cycles, the discharge and charge capacity are both 0.41 mAh cm<sup>-2</sup>, showing no obvious capacity attenuation. Besides, even after 140 cycles at various rates from 0.3 C to

60 C, the discharge capacity of the Cu<sub>2</sub>S/Cu cathode can restore 97.8% when lowering the charge/discharge rate to 0.3 C. The excellent rate capability and cycle stability of Cu<sub>2</sub>S/Cu cathode suggest that metal sulfides have potential application in high-rate Li-ion batteries, and improving of initial coulombic efficiency may be an important issue to be resolved.

One of the serious problems of the Li/CuS batteries is a rapid drop in capacity during cycling [85, 86] which is caused by the high solubility (in organic solvent electrolytes) of the  $\text{Li}_2\text{S}$  that are formed during the charge/discharge processes [87]. The dissolution of the products in the charge-discharge processes can lead to low active materials utilization, low coulombic efficiency and short cycle life of the CuS cathode [88]. In order to solve these problems, the two approaches were employed for improving the cycle life of CuS cathode. One is the replacement of liquid electrolyte with inorganic solid electrolyte [89]; the other is limiting the discharge cut-off voltage to avoid the formation of  $\text{Li}_2\text{S}$ .

Yourong Wang et al. [87] reported for the first time, the in situ preparation of the CuS cathode on the Cu foil substrate for lithium ion battery by a facile method, and the electrochemical properties of the obtained CuS cathode were investigated under different current density. The possible mechanism of the unique stability of the CuS cathode was discussed. Hierarchical-structured copper sulfide/multi-walled carbon nanotubes (CuS/MWCNTs) are synthesized via a one-step hydrothermal process [90]. As expected, the CuS-MWCNTs exhibit a much higher specific capacitance up to  $2831 \text{ F g}^{-1}$ , compared with

$925.1 \text{ F g}^{-1}$  for CuS and  $555.6 \text{ F g}^{-1}$  for MWCNTs. Furthermore, the CuS-MWCNTs hybrids also exhibit good cycling stability with more than 90% capacitance retention over 600 cycles. The enhancement of CuS/MWCNTs in supercapacitor performance not only attribute to their unique 3D structures with large specific surface area, but also their excellent conductivity, which facilitate efficient charge transport and promotes electrolyte diffusion.

An oriented attachment and growth mechanism allows an accurate control of the size and morphology of  $\text{Cu}_{2-x}\text{S}$  nanocrystals, from spheres and disks to tetradecahedrons and dodecahedrons [91]. The performance of  $\text{Cu}_{2-x}\text{S}$  nanocrystals as a cathode in all-vanadium redox flow batteries (VRB) was also tested. The efficiency of this type of battery is usually limited by the rate and potential of the  $[\text{VO}]^{2+}/[\text{VO}_2]^+$  cathodic reaction.  $\text{Cu}_{2-x}\text{S}$  nanocrystals were thoroughly purified, deposited on a substrate and fixed by Nafion. Their electrochemical activity was characterized in an inert atmosphere by means of cyclic voltammetry. Their characteristics were compared with those obtained for a polyacrylonitrile (PAN)-derived graphite felt, which is a material widely used as a VRB cathode. Lower oxidation potentials were systematically obtained for  $\text{Cu}_{2-x}\text{S}$  nanocrystals when compared to the PAN-based graphite felt. At a  $2 \text{ mV s}^{-1}$  scan rate, the oxidation potential was found at  $0.72 \text{ V}$  for the PAN-based graphite felt and at  $0.44 \text{ V}$  for the electrode containing  $\text{Cu}_{2-x}\text{S}$  nanocrystals. This result denotes faster electrocatalytic kinetics of the oxidation process for the electrode containing  $\text{Cu}_{2-x}\text{S}$  nanocrystals. The difference of potential between the oxidation and reduction peaks was  $0.36 \text{ V}$  for the PAN-based graphite felt and  $0.10 \text{ V}$  for the  $\text{Cu}_{2-x}\text{S}$  electrode. Furthermore, the ratio between the currents at the oxidation and reduction peaks for  $\text{Cu}_{2-x}\text{S}$  nanocrystals was close to unity and changed moderately with the scan rate. These experimental results pointed towards a significant performance improvement in terms of reversibility of the  $[\text{VO}]^{2+}/[\text{VO}_2]^+$  redox process with the use of  $\text{Cu}_{2-x}\text{S}$  nanocrystals [91]. However, the currents measured with  $\text{Cu}_{2-x}\text{S}$  cathodes were lower than those obtained with PAN-based graphite felts.

**2.1.2. Fuel cells.** As devices for directly converting chemical energy into electrical energy, fuel cells that oxidize fuel at the anode and reduce oxygen from the air at the cathode to produce electricity for automotive propulsion have achieved great progress in many technical aspects related to their large-scale practical applications [92]. Efficient catalysts for the oxygen reduction reaction (ORR;  $\text{O}_2 + 4\text{H}^+ + 4\text{e}^- \rightarrow 2\text{H}_2\text{O}$ ) are essential for the fabrication of high performance direct methanol fuel cells (DMFCs) [92]. Many nanomaterials (NMs) such as nitrogen-doped carbon [93,94], CuS [95,96] have been prepared and employed as effective catalysts for high-rate ORR kinetics by taking advantage of their large surface area and high catalytic activities [97]. Low-cost MS (M = Co and Cu) exhibits metal-like electrical conductivity, a property that has attracted significant interest in electrochemistry; however, current MS electrodes have much lower ORR activities than Pt electrodes [7, 98]. Carbon supports are commonly used in fuel cells to improve the electrochemical activity and stability of catalysts mainly because of their large surface area, low cost, and unique electrical properties [99]. Carbon-supported  $\text{CoS}_x$  and  $\text{CoSe}_x$  NMswith onset potential values of  $0.67 \text{ V}$  and  $0.72 \text{ V}$ , respectively, are more active in the ORR than a reversible hydrogen electrode in acidic media [7, 98].

**2.1.3. Solar cells.** CuS nanoparticles or nanowires could also be grown on carbon or Cu substrates, and the nano devices fabricated using binding of  $\text{Cu}_2\text{S}$  with the conductive substrates may lead to more excellent performance in solar cells and the sensors [50]. In the copper sulfide family, there are several known solid phases like  $\text{Cu}_2\text{S}$  ( $\beta$ ,  $\gamma$  -chalcocite),  $\text{Cu}_{1.96}\text{S}$  (djurite),  $\text{Cu}_{1.8}\text{S}$  (digenite), and CuS (covellite). All of these phases have been identified as p-type semiconducting materials because of the copper vacancies within the lattice [49]. Among them,  $\text{Cu}_2\text{S}$  is both an indirect and direct band gap materials, with  $E_{g,\text{ind}} \approx 1.2 \text{ eV}$  and  $1.8 \text{ eV}$ , respectively [100,101], and considered as an ideal absorber in photovoltaic conversions due to its high absorption coefficient ( $10^4 \text{ cm}^{-1}$ ) [100] has been widely used in solar cell devices.

$\text{Cu}_2\text{S}$  nanocrystalline films have been successfully prepared using dip-coating technique [102]. The absorption coefficient of  $\text{Cu}_2\text{S}$  film in the UV-vis spectrum of the  $\text{Cu}_2\text{S}$ , exhibits a high absorption characteristic. The absorption coefficient of spectrum is in the range  $10^4 \text{ cm}^{-1}$ – $10^5 \text{ cm}^{-1}$ . This feature is desirable for the photovoltaic devices applications. Furthermore, it is attractive as a selective solar absorber since it has a high solar absorbency and a low thermal emission [103].

Jeunghhee Park et al. fabricated the solar cells and the amperometric glucose sensors using these Cu<sub>2</sub>S–MWCNT hybrid nanostructures respond more sensitively than those using the Cu<sub>2</sub>S NCs (or MWCNTs) alone. [50,59] The solar cells and the amperometric glucose sensors fabricated using these Cu<sub>2</sub>S–MWCNT hybrid nanostructures respond more sensitively than those using the Cu<sub>2</sub>S NCs (or MWCNTs) alone. The utilization of the active Cu<sub>2</sub>S NCs through direct binding with the conductive MWCNTs would lead to excellent performance of these nanodevices [50]. Su-Huai Wei et al. [104] find that under Cu-rich condition, the anilite Cu<sub>11</sub>3S<sub>8</sub> is the most stable structure. It has a predicted band gap of 1.4 eV and could be a promising solar cell absorber. With a direct energy bandgap of 1.5 eV approaching the most appropriate range for high photovoltaic conversion and a high absorption coefficient of more than 10<sup>4</sup> cm<sup>-1</sup>, CuInS has been regarded as a promising candidate for thin film solar cell applications [105]. Solar energy conversion efficiencies above 10% have been obtained with CuInS devices [106]. Using these thin films for the absorption layer of solar cells have attracted growing attention from poly- crystalline ternary semiconductors [107] due to its wide band gap, high absorption coefficient, broad spectrum absorption, and high thermal stability [108].

**3. Summary and perspective.** In this review, we aim to provide a summary on the properties and perspective applications of nanostructured copper sulfide materials. Controlling copper sulfide nanocrystals with an appropriate crystal size is also important. Bigger crystal sizes will lead to smaller specific surface areas and thus lower activity, while smaller sizes often suffer from faster collapse in the crystal-line structure and thus lower stability.

Recent advances have proved that nanostructured copper sulfide materials are great potential materials for energy conversion and storage utilization because of their unique physical and chemical properties and because of the above listed “nanoscale effects”. The huge energy requirement and fossil-fuel induced environmental pollution create an enormous strain for scientists to develop clean and sustainable technologies to provide abundant energy in an economically viable way. Energy conversion and storage devices such as fuel cells, photoelectrochemical water splitting cells, solar cells, Li-ion batteries and supercapacitors, etc. have the capability to power the energy demanding areas that range from portable electronics to transportation and even stationary.

## References

1. Lindroos S, Arnold A, Leskela M. Growth of CuS thin films by the successive ionic layer adsorption and reaction method // *Applied Surface Science*. 2000. V. 158, P. 75–80.
2. Lin M.Ch, Lee M.V. Cu<sub>2</sub>S quantum dot-sensitized solar cells // *Electrochemistry Communications*. 2011, V. 13, P. 1376–1378.
3. Joseph M. Luther, Prashant K. Jain, Trevor Ewers. A. Paul Alivisatos. Localized surface plasmon resonances arising from free carriers in doped quantum dots // *Nature Materials*. 2011, V. 10, P. 361–366.
4. Qian L, Mao J, Tian X, Yuan H, Xiao D. In situ synthesis of CuS nanotubes on Cu electrode for sensitive nonenzymatic glucose sensor // *Sensors and Actuators B*. 2013, V. 176, P. 952–959.
5. Wu Ch, Yu Sh. H., Chen Sh, Liu G, Liu B. Large scale synthesis of uniform CuS nanotubes in ethylene glycol by a sacrificial templating method under mild conditions // *J. Mater. Chem*. 2006, V. 16, P. 3326–333.
6. Roy P, Srivastava S. K. Hydrothermal growth of CuS wires from Cu-dithiooxamide, a novel single source precursor // *Crystal Growth and Design*. 2006, V. 6, P. 1921–1926
7. Lai C.H, Huang K.W, Cheng J.H, Lee C.Y, Hwang B.J, Chen L.J. Direct growth of high-rate capability and high capacity copper sulfide nanowire array cathodes for lithium-ion batteries // *J. Mater. Chem*. 2010, V. 20, P. 6638–6645.
8. Yu X, An X. Controllable hydrothermal synthesis of CuS nanowires on the copper substrate // *Materials Letters*. 2010, V. 64, P. 252–254.
9. [9]. Roy P, Mondal K, Srivastava S.K. Synthesis of twinned CuS nanorods by simple wet chemical method // *Crystal Growth and Design*. 2008, V. 8, P. 1530–1534.
10. Roy P, Srivastava S.K. Low-temperature synthesis of CuS nanorods by simple wet chemical method // *Mater. Lett*. 2007, V. 61, P. 1693–1697.
11. Jiang X, Xie Y, Lu J, He W, Zhu L, Qian Y. Preparation and phase transformation of nanocrystalline copper sulfides (Cu<sub>2</sub>S, Cu<sub>2</sub>Se and CuS) at low temperature // *J. Mater. Chem*. 2000, V. 10, P. 2193.
12. Zhu T, Xia B.Y, Zhou L, Lou X. W. Arrays of ultrafine CuS nanoneedles supported on a CNT backbone for application in supercapacitors // *J. Mater. Chem*. 2012, V. 22, P. 7851–7855.
13. Kundu J, Pradhan D. Influence of precursor concentration, surfactant and temperature on the hydrothermal synthesis of CuS: structural, thermal and optical properties // *New J. Chem*. 2013, V. 37, P. 1470–1478.
14. Lim W.P, Wong C.T, Ang S.L, Low H.Y, Chin W.S. Phase-Selective Synthesis of Copper Sulfide Nanocrystals // *Chemistry of Materials*. 2006, V. 18, P. 6170–6177.
15. Xu H, Wang W, Zhu W. Sonochemical synthesis of crystalline CuS nanoplates via an in situ template route // *Materials Letters*. 2006, V. 60, P. 2203–2206.
16. Ji R, Wang L, Wang G, Zhang X. Synthesize thickness copper (I) sulfide nanoplates on copper rod and its application as nonenzymatic cholesterol sensor // *Electrochimica Acta*. 2014, V. 130, P. 239–244.



17. Zhang H, Zhang Y, Yu J, Yang D. Phase-selective synthesis and self-assembly of monodisperse copper sulfide nanocrystals // *Journal of Physical Chemistry C*. 2008, V. 112, P. 13390–13394.
18. Zhang J, Yu J, Zhang Y, Li Q, Gong J.R. Visible light photocatalytic H<sub>2</sub>-production activity of CuS/ZnS porous nanosheets based on photoinduced interfacial charge transfer // *Nano Lett.* 2011, V. 11, P. 4774–4779.
19. Xua L, Chena X, Mab L, Gao F. Facile preparation of copper sulfide nanoparticles from perovskite templates containing bromide anions // *Colloids Surf. A*. 2009, V. 349, P. 69–73.
20. Li Z, Fangqi T, Zhao Q, Xianlong Z, Yuhong Y, Juncheng H. Bubble template synthesis of copper sulfide hollow spheres and their applications in lithium ion battery // *Materials Letters*. 2012, V. 68, P. 28–31.
21. Nagarathinam M, Saravanan K, Leong W.L, Balaya P, Vittal J.J. Hollow nanospheres and flowers of CuS from self-assembled Cu(II) coordination polymer and hydrogen-bonded complexes of N-(2-Hydroxybenzyl)-l-serine. *Cryst Growth Des* 2009, V. 9, P. 4461–4470.
22. Z.H. Yang, D.P. Zhang, W.X. Zhang, M. Chen, Controlled synthesis of cuprous oxide nanospheres and copper sulfide hollow nanospheres. *J. Phys. Chem. Solids* 70 (2009) 840–846.
23. Y. Wang, Y. Hu, Q. Zhang, J. Ge, Z. Lu, Y. Hou, Y. Yin, One-Pot Synthesis and Optical Property of Copper(I) Sulfide Nanodisks. *Inorganic Chemistry* 49 (2010) 6601–6608.
24. Shaodong S, Zhimao Yang. Cu<sub>2</sub>O-templated strategy for synthesis of definable hollow architectures. *Chem. Commun.*, 2014, 50, 7403–7415.
25. Jiao S, Xu L, Jiang K, Xu D. Well-defined non-spherical copper sulfide mesocages with single-crystalline shells by shape-controlled Cu<sub>2</sub>O crystal templating // *Adv. Mater.* 2006, V. 18, P. 1174–1177.
26. Ramli E, Rauchfuss T.B, Stern C.L. Interception of copper polysulfide clusters in the reaction of copper and sulfur in donor solvents: polysulfide complexes as the link between molecular and non-molecular metal sulfides // *J. Am. Chem. Soc.* 1990, V. 112, P. 4043–4044.
27. Motomichi I, Catalina C.V, Michiko B.I. Quintus F. Electroconducting transparent film of amorphous copper sulfide on polyethylene substrate // *J. Mater. Chem.*, 1992, V. 2, P. 761–762.
28. Anuar K, Zainal Z, Hussein M.Z, Saravanan N, Haslina I. Cathodic electrodeposition of Cu<sub>2</sub>S thin film for solar energy conversion // *Sol. Energy Mater. Sol. Cells*. 2002, V. 73, P. 351–365.
29. Dinsmore A.D, Hsu M.F, Nikolaides M.G, Marquez M, Bausch A.R. Weitz DA. Colloidosomes: selectively permeable capsules composed of colloidal particles // *Science* 2002, V. 298, P. 1006–1009.
30. Bigi A, Boanini E, Walsh D, Mann S. Morphosynthesis of octacalcium phosphate hollow microspheres by polyelectrolyte-mediated crystallization // *Angew Chem Int Ed* 2002, V. 41, P. 2163–2166.[31].
31. Liu B, Zeng H.C. Mesoscale organization of CuO nanoribbons: formation of “Dandelions” // *J Am Chem Soc* 2004, V. 126, P. 8124–8125.
32. Nakai I, Sugitani T, Nagashima K, Niwa Y. X-ray photoelectron spectroscopic study of copper minerals // *J. Inorg. Nucl. Chem.* 1978, V. 40, P. 789.
33. Folmer J.C.W, Jellinek F. The valence of copper in sulphides and selenides: An X-ray photoelectron spectroscopy study // *J. Less Common metals* 1980, V.76, P. 153.
34. Westrum E.F, Stolen S, Gronvold F. Thermodynamics of copper sulfides. 2. Heat-capacity and thermodynamic properties of synthetic covellite, CuS, from 5-K to 780.5-K—Enthalpy of decomposition // *Journal of Chemical Thermodynamics*. 1987, V. 19, P. 1199–1208.
35. Potter R.W. An electrochemical investigation of the system copper-sulfur // *Econ. Geol.* 1977, V. 72, P. 1524.
36. Okamoto K, Kawai S, Kiriya M, R. Jap. Electrical and Magnetic Properties of Cu<sub>3</sub>Se<sub>2</sub> and Some Related Compounds // *J. Appl. Phys.* 1969, V. 8, P. 718–724.
37. An C, Wang S, He J, Wang Z. A composite-surfactants-assisted-solvothermal process to copper sulfide nanocrystals // *J. Cryst. Growth*. 2008, V. 310, P. 266–269. <http://dx.doi.org/10.1016/j.jcrysgro.2007.11.017>
38. Sheng W, Kim S, Lee J, Kim S.W, Jensen K, Bawendi M.G. In-situ encapsulation of quantum dots into polymer microspheres // *Langmuir*. 2006, V. 22, P. 3782–3790.
39. Orphanou M, Leontidis E, Leodidou T.K, Koutsoukos P, Kyriacou K.C. Study of copper sulfide crystallization in PEO–SDS solutions // *Langmuir*. 2004, V. 20, P. 5605–5612.
40. Klimov V, Bolivar P.H, Kurz H, Karavanskii V, Korkishko Y. Linear and nonlinear transmission of Cu<sub>2</sub>S quantum dots // *Appl. Phys. Lett.* 1995, V. 67, P. 653–655.
41. Yin Y, Alivisatos A. P. Colloidal nanocrystal synthesis and the organic-inorganic interface // *Nature* 2005, V.437, P. 664–670.
42. Wei X, Kong T, Zhu H, Wang L. CuS/Cu<sub>2</sub>S nanofluids: Synthesis and thermal conductivity // *International Journal of Heat and Mass Transfer*. 2010, V. 53, P. 1841–1843.
43. Hsu Y.K, Chen Y.C, Lin Y.G, Chen L.C, Chen K.H. High-cell-voltage supercapacitor of carbon nanotube/carbon cloth operating in neutral aqueous solution // *J. Mater. Chem.* 2012, V. 22, P. 3383–3387.
44. Hsu Y.K, Chen Y.C, Lin Y.G. Synthesis of copper sulfide nanowire arrays for high-performance supercapacitors // *Electrochimica Acta* 2014, V. 139, P. 401–407.

45. Yamakawa N, Jiang M, Grey C.P. Investigation of the conversion reaction mechanisms for binary copper(II) compounds by solid-state NMR spectroscopy and X-ray diffraction // *Chem Mater.* 2009, V. 21, P. 3162–3176.
46. Mane R.S, Lokhande C.D. Chemical deposition method for metal chalcogenide thin films // *Mater. Chem. Phys.* 2000, V. 65, P. 1–31.
47. Erokhina S, Erokhin V, Nicolini C. Microstructure origin of the conductivity differences in aggregated CuS films of different thickness // *Langmuir.* 2003, V. 19, P. 766–771.
48. Blachnik R, Müller A. The formation of Cu<sub>2</sub>S from the elements: I. Copper used in form of powders // *Thermochim. Acta* 2000, V. 361, P. 31–52.
49. Mukherjee N, Sinha A, Khan G.G, Chandra D, Bhaumik A, Mondal A. A study on the structural and mechanical properties of nanocrystalline CuS thin films grown by chemical bath deposition technique // *Materials Research Bulletin.* 2011, V. 46, P. 6–11.
50. Wu Y, Wadia C, Ma W, Sadtler B, Alivisatos A.P. Synthesis and photovoltaic application of copper (I) sulfide nanocrystals // *Nano Lett.* 2008, V. 8, P. 2551–2555.
51. Lee H, Yoon S.W, Kim E.J, Park J. In situ growth of copper sulfide nanocrystals on multi-walled carbon nanotubes and their application as novel solar cell and amperometric glucose sensor materials // *Nano Lett.* 2007, V. 7, P. 778–784.
52. Johansson J, Kostamo J, Karppinen M, Niinisto L. Growth of conductive copper sulfide thin films by atomic layer deposition // *J. Mater. Chem.* 2002, V. 12, P. 1022–1026.
53. Grozdanov I, Najdoski M. Optical and electrical properties of copper sulfide films of variable composition // *J. Solid State Chem.* 1995, V. 114, P. 469–475.
54. Li L, Zhang Y, Fang X.S, Zhai T.Y, Liao M.Y, Sun X.L, Koide Y, Bando Y, Golberg D. WO<sub>3</sub> nanowires on carbon papers: electronic transport, improved ultraviolet-light photodetectors and excellent field emitters // *J. Mater. Chem.* 2011, V. 21, P. 6525–6530.
55. Lai C.H, Lu M.Y, Chen L. J. Metal sulfide nanostructures: synthesis, properties and applications in energy conversion and storage // *J. Mater. Chem.* 2012, V. 22, P. 19–30.
56. Zhao B, Li Sh, Zhang Q, Wang Y, Song Ch, Zhang Z, Yu K. Controlled synthesis of Cu<sub>2</sub>S microrings and their photocatalytic and field emission properties // *Chemical Engineering Journal.* 2013, V. 230, P. 236–243.
57. Li S, Yu K, Wang Y, Zhang Z, Song C, Yin H. Cu<sub>2</sub>S@ZnO hetero-nanostructures: facile synthesis, morphology-evolution and enhanced photocatalysis and field emission properties // *Cryst Eng Comm* 2013, V. 15, P. 1753–1761.
58. Song Ch, Yin H, Zhang N, Li Sh, Zhao B, Yu K. Two-step synthesis of novel Cu<sub>2</sub>S nanoflowers for field emission application // *Materials Letters.* 2014, V. 137, P. 56–58.
59. Li L, Fang X, Chew H.G, Zheng F, Liew T.H, Xu X. Crystallinity-controlled germanium nanowire arrays: Potential field emitters // *Adv Funct Mater* 2008, V. 18, P. 1080–1088.
60. Song Ch, Yu K, Li Sh, Yin H, Zhang N, Zhao B, Zhu Z. Controlled synthesis of novel rod-like Cu<sub>1.8</sub>S nanostructures and field emission properties // *Applied Surface Science.* 2014, V. 315, P. 235–240.
61. Yu K, Zhang Y.S, Xu F, Li Q, Zhu Z.Q, Wan Q. Significant improvement of field emission by depositing zinc oxide nanostructures on screen-printed carbon nanotube films // *Appl. Phys. Lett.* 2006, V. 88, P. 153123.
62. Gandubert V. J, Torres E, Niemeyer C. M. Investigation of cytochrome P450-modified cadmium sulfide quantum dots as photocatalysts // *J. Mater. Chem.* 2008, V. 18, P. 3824–3830.
63. Lu M. Y, Lu M. P, Chung Y. A, Chen M. J, Wang Z. L, Chen L. J. Intercrossed sheet-like Ga-doped ZnS nanostructures with superb photocatalytic activity and photoresponse // *J. Phys. Chem. C.* 2009, V. 113, P. 12878–12882.
64. Li F, Wu J, Qin Q, Li Z, Huang X. Controllable synthesis, optical and photocatalytic properties of CuS nanomaterials with hierarchical structures // *Powder Technology.* 2010, V. 198, P. 267–274.
65. Zhang Z, Wang C.C, Zakaria R, Ying J.Y. Role of particle size in nanocrystalline TiO<sub>2</sub>-based photocatalysts // *J. Phys. Chem. B.* 1998, V. 102, P. 10871–10878.
66. [65]. Ding T.Y, Wang M.S, Guo S.P, Guo G.C, Huang J.S. CuS nanoflowers prepared by a polyol route and their photocatalytic property // *Mater. Lett.* 2008, V. 62, P. 4529–4531.
67. Saranya M, Santhosh Ch, Ramachandran R, Kollu P, Saravanan P, Vinoba M, Jeong S.K, Grace A.N. Hydrothermal growth of CuS nanostructures and its photocatalytic properties // *Powder Technology.* 2014, V. 252, P. 25–32.
68. Thuy U.T.D, Liem N.Q, Parlett C.M.A, Lalev G.M, Wilson K. Synthesis of CuS and CuS/ZnS core/shell nanocrystals for photocatalytic degradation of dyes under visible light // *Catalysis Communications.* 2014, V. 44, P. 62–67.
69. Dittmer W.U, Simmel F.C. Chains of semiconductor nanoparticles templated on DNA // *Appl. Phys. Lett.* 2004, V. 85, P. 633–635.
70. [Savariraj A.D, Viswanathan K.K, Prabakar K. CuS nano flakes and nano platelets as counterelectrode for quantum dots sensitized solar cells // *Electrochimica Acta.* 2014, V. 149, P. 364–369.
71. Raj C.J, Kim B.C, Cho W.J, W.G, Seo Y, Yu K.H. Electrochemical capacitor behavior of copper sulfide (CuS) nanoplatelets // *Journal of Alloys and Compounds* 2014, V. 586, P. 191–196.
72. Orphanou M, Leontidis E, Leodidou T.K, Koutsoukos P, Kyriacou K.C. Study of Copper Sulfide Crystallization in PEO-SDS Solutions // *Langmuir.* 2004, V. 20, P. 5605–5612.

73. Sartale S.D, Lokhande C.D. Growth of copper sulphide thin films by successive ionic layer adsorption and reaction (SILAR) method // *Mater Chem Phys.* 2000, V. 65, P. 63-67.
74. Bordiga S, Paze C, Berlier G, Scarano D, Spoto G, Zecchina A, et al. Interaction of N<sub>2</sub>, CO and NO with Cu-exchanged ETS-10: A compared FTIR study with other Cu- zeolites and with dispersed Cu<sub>2</sub>O // *Catal Today* 2001, V. 70, P. 91-105.
75. [74]. Xie Y, Li Y, Niu L, Wang L, Qian H, Yao W. A novel surface-enhanced Raman scattering sensor to detect prohibited colorants in food by graphene/silver nanocomposite // *Talanta.* 2012, V. 100, P. 32-37.
76. Khodadoust S, Ghaedi M, Sahraei R, Daneshfar A. Application of experimental design for removal of sunset yellow by copper sulfide nanoparticles loaded on activated carbon // *Journal of Industrial and Engineering Chemistry.* 2014, V. 20, P. 2663-2670.
77. Ghaedi M, Khodadoust S, Sadeghi H, Khodadoust M. A, Armand R, Fatehi A. Application of ultrasonic radiation for simultaneous removal of auramine O and safranin O by copper sulfide nanoparticles: Experimental design // *Spectrochimica Acta Part A: Molecular and Biomolecular Spectroscopy.* 2015, V. 136, P. 1069-1075.
78. Cai X.J, Gao X, Wang L.S, Wu Q, Lin X.F. A layer-by-layer assembled and carbon nanotubes/gold nanoparticles-based bienzyme biosensor for cholesterol detection // *Sens. Actuators. B.* 2013, V. 181, P. 575-583.
79. Duan M, Peng Y.L, Zhang L.L, Wang X.Y, Ge J, Jiang J.H, Yu R.Q. DNA-stabilized silver nanoclusters with guanine-enhanced fluorescence as a novel indicator for enzymatic detection of cholesterol // *Anal. Methods.* 2013, V. 5, P. 2182-2187.
80. Park S, Boo H, Chung T.D. Electrochemical non-enzymatic glucose sensors. // *Anal. Chim.*
81. *Acta.* 2006, V. 556, P.46-57.
82. Pickup J.C, Hussain F, Evans N.D, Sachedina N. In vivo glucose monitoring: the clinical reality and the promise // *Biosensors & Bioelectronics.* 2005, V.20, P. 1897-1902.
83. Mayorga-Martinez C.C, Guix M, Madrid R.E, Merkoci A. Bimetallic nanowires as electrocatalysts for nonenzymatic real-time impedancimetric detection of glucose // *Chemical Communications.* 2012, V. 48, P. 1686-1688.
84. Liu J, Xue D.F. Rapid and scalable route to CuS biosensors: a microwave assisted Cu-complex transformation into CuS nanotubes for ultrasensitive nonenzymatic glucose sensor // *Journal of Materials Chemistry.* 2011, V. 21, P. 223-228.
85. Su X, Wu Q, Zhan X, Wu J, Wei S, Guo Z. Advanced titania nanostructures and composites for lithium ion battery // *Journal of Materials Science.* 2012, V. 47, P. 2519-2534.
86. Ni Sh, Lv X, Li T, Yang X. Fabrication of Cu<sub>2</sub>S cathode for Li-ion battery via a low temperature dry thermal sulfuration method // *Materials Chemistry and Physics* 2013, V. 143, P. 349-354.
87. Exner I, Hep J. Copper (III) sulfide as cathode active material in secondary lithium batteries // *Journal of Power Sources.* 1993, V. 44, P. 701-705.
88. Han Y, Wang Y, Gao W. Synthesis of novel CuS with hierarchical structures and its application in lithium-ion batteries // *Powder Technology.* 2011, V. 212, P. 64-68.
89. Wang Y, Zhang, Chen X.P, Liao H, Cheng S. In situ preparation of CuS cathode with unique stability and high rate performance for lithium ion batteries // *Electrochimica Acta.* 2012, V. 80, P. 264- 268.
90. Ji X, Evers S, Black R, Nazar L.F. Stabilizing lithium-sulphur cathodes using polysulphide reservoirs // *Nature Communications.* 2011, V. 2, P. 325-331.
91. Yan J.M, Huang H.Z, Zhang J, Liu Z.J, Yang Y. A study of novel anode material CoS<sub>2</sub> for lithium ion battery // *Journal of Power Sources.* 2005, V. 146, P. 264-269.
92. Huang K.J, Zhang J.Z, Xing K. One-step synthesis of layered CuS/multi-walled carbon nanotube nanocomposites for supercapacitor electrode material with ultrahigh specific capacitance // *Electrochimica Acta.* 2014, V. 149, P. 28-33.
93. Li W, Shavel A, Guzman R, Rubio-Garcia J, Flox C, Fan J, Cadavid D, Ibanez M, Arbiol J, Morante JR, Cabot A. Morphology evolution of Cu<sub>2</sub>S nanoparticles: from spheres to dodecahedrons // *Chem. Commun.* 2011, V. 47, P. 10332-10334.
94. Gewirth A.A, Thorum M.S. Electroreduction of dioxygen for fuel-cell applications: materials and challenges // *Inorganic Chemistry.* 2010, V. 49, P. 3557-3566.
95. Shanmugam S, Osaka T. Efficient electrocatalytic oxygen reduction over metal free-nitrogen doped carbon nanocapsules // *Chemical Communications.* 2011, V. 47, P. 4463-4465.
96. Morozan A, Josselme B, Palacin S. Low-platinum and platinum-free catalysts for the oxygen reduction reaction at fuel cell cathodes // *Energy Environment Science* 2011, V. 4, P. 1238- 1254.
97. Wang Y, Li Q, Nie M, Li X, Li Y, Zhong X. [High-yield room temperature route to copper sulfide hollow nanospheres and their electrochemical properties](#) // *Nanotechnology.* 2011, V. 22, P. 305401-305407.
98. Feng Y, He T, Alonso-Vante N. In situ free-surfactant synthesis and ORR- electrochemistry of carbon-supported Co<sub>3</sub>S<sub>4</sub> and CoSe<sub>2</sub> nanoparticles // *Chemistry of Materials.* 2008, V. 20, P. 26-28.
99. Steele B.C.H, Heinzel A. Materials for fuel-cell technologies // *Nature.* 2001, V. 414, P. 345-100. 352.
101. Wang H, Liang Y, Li Y, Dai H. Co<sub>0.40</sub>S-Graphene hybrid: A high-performance metal chalcogenide electrocatalyst for oxygen reduction // *Angewandte Chemie International Edition* 2011, V. 50, P. 10969-10972.

102. Antolini E. Platinum-based ternary catalysts for low temperature fuel cells: Part I. Preparation methods and structural characteristics // *Applied Catalysis B: Environmental* 2007, V. 74, P. 324–336.
103. Liu G, Schulmeyer T, Brotz J, Klein A, Jaegermann W. Interface properties and band alignment of Cu<sub>2</sub>S/CdS thin film solar cells // *Thin solid films* 2003, V. 431-432, P. 477-482.
104. He Y, Kriegseis W, Bläsing J, Polity A, Krämer T, Hasselkamp D, Meyer BK, Hardt M, Krost A. (001)-Textured Cu<sub>2</sub>S thin films deposited by RF reactive sputtering // *J. Appl. Phys.* 2002, V.41, P. 4630-4634.
105. Rafea M.A, Farag A.A.M, Roushdy N. Controlling the crystallite size and influence of the film thickness on the optical and electrical characteristics of nanocrystalline Cu<sub>2</sub>S films // *Materials Research Bulletin*. 2012, V. 47, P. 257–266.
106. Wemple S.H, DiDomenico M. Theory of the elasto-optic effect in nonmetallic crystals // *J. Phys. Rev. B*. 1970, V. 1, P. 193-202.
107. Xu Q, Huang B, Zhao Y, Yan Y, Noufi R, Wei S.H. Crystal and electronic structures of Cu<sub>2</sub>S solar cell absorbers // *Appl. Phys. Lett.* 2012, V. 100, P. 061906. [105] Alt M, Lewerenz H.J, Scheer R. CuInS<sub>2</sub> thin film growth monitoring by in situ electric conductivity measurements // *J. Appl. Phys* 1997, V. 81, P. 956-961.
108. Dzionk C, Metzner H, Hessler S, Mahnke H.E. Phase formation during the reactive annealing of Cu–In films in H<sub>2</sub>S atmosphere // *Thin Solid Films*. 1997, V. 299, P. 38-44.
109. Cho JW, Park SJ, Kim W, Min BK, Fabrication of nanocrystal ink based superstrate-type CuInS<sub>2</sub> thin film solar cells, *Nanotechnology* 2012,23,265401.
110. Chen H, Yeh Y.M, Liao C.H, Chen J.Z, Wang C. Removal of CuS phases from electrodeposited CuInS<sub>2</sub> films // *Ceramics International* 2014, V. 40, P. 67–72.
111. Wang Y. J, Wilkinson D. P, Zhang J. J. Noncarbon support materials for polymer electrolyte membrane fuel cell electrocatalysts // *Chem. Rev.*, 2011, V. 111, P. 7625–7651.
112. Gao M.R, Xu Y.F, Jiang J, Yu Sh.H. Nanostructured metal chalcogenides: synthesis, modification, and applications in energy conversion and storage devices // *Chem. Soc. Rev.* 2013, V. 42, P. 2986-3017.